

# Surface Chemistry, Electronic Structure, and Activity of SOFC Cathode Films

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July 25, 2012



# Solid Oxide Fuel Cells Cathode Materials

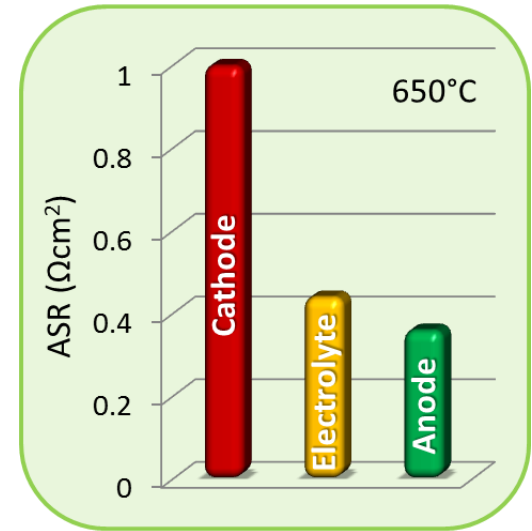
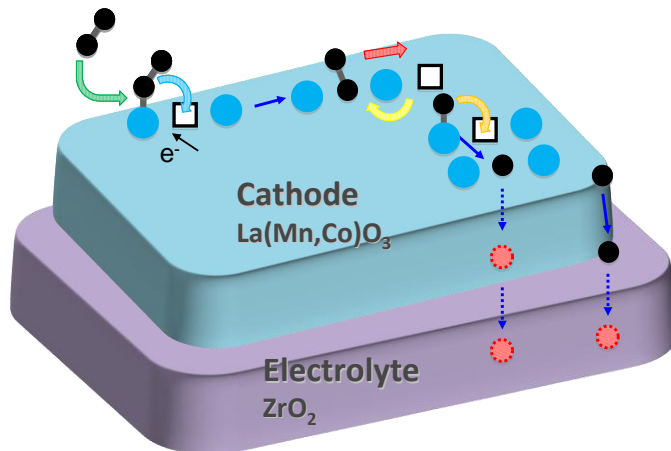
- ❑ SOFC *cathode* materials
  - : Main barrier to achieve higher power
- ❑ Oxygen Reduction (OR) at SOFC cathode

Structure

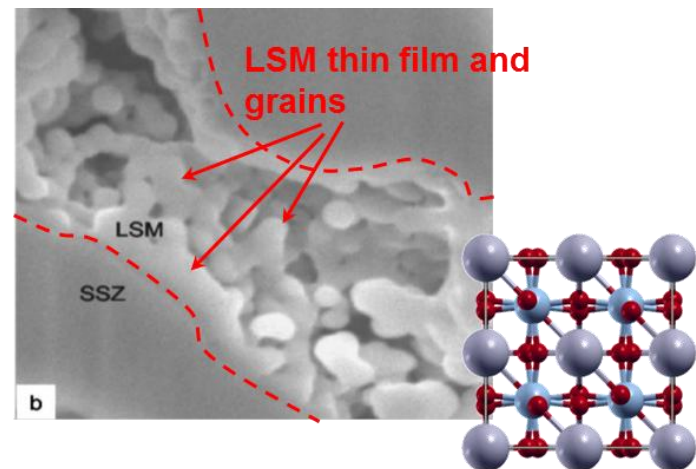
Composition

Electronic structure

↔ OR kinetic



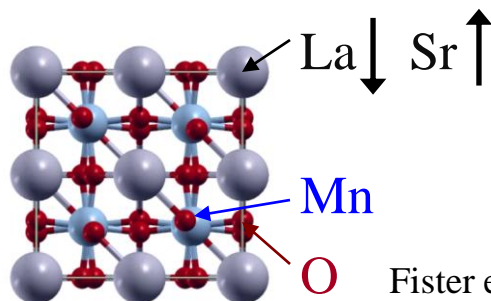
Singhal and Kendall, *High Temperature Solid Oxide Fuel Cells*, 2003.



Sholklapper et al., *Electrochem. Solid. St.* **10** (2007).

# Surface Structure and Chemistry: A-Site Rich?

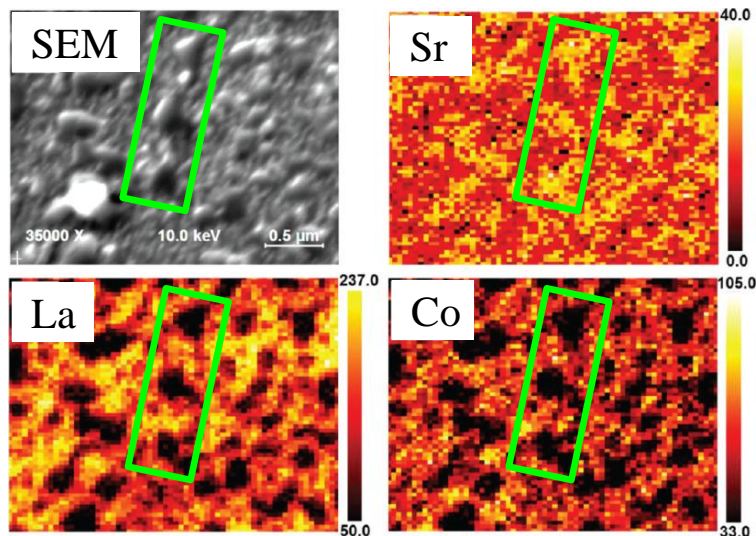
□ Sr rich in A-site [1]



Fister et al., *Appl. Phys. Lett.* **93** (2008).

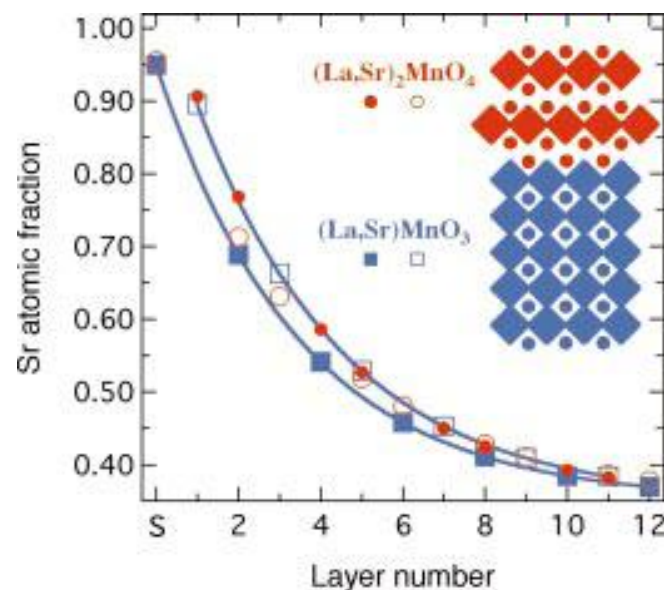
□ Formation of (La,Sr)O [2]

(La,Sr)CoO<sub>3</sub> annealed at 650 C in air



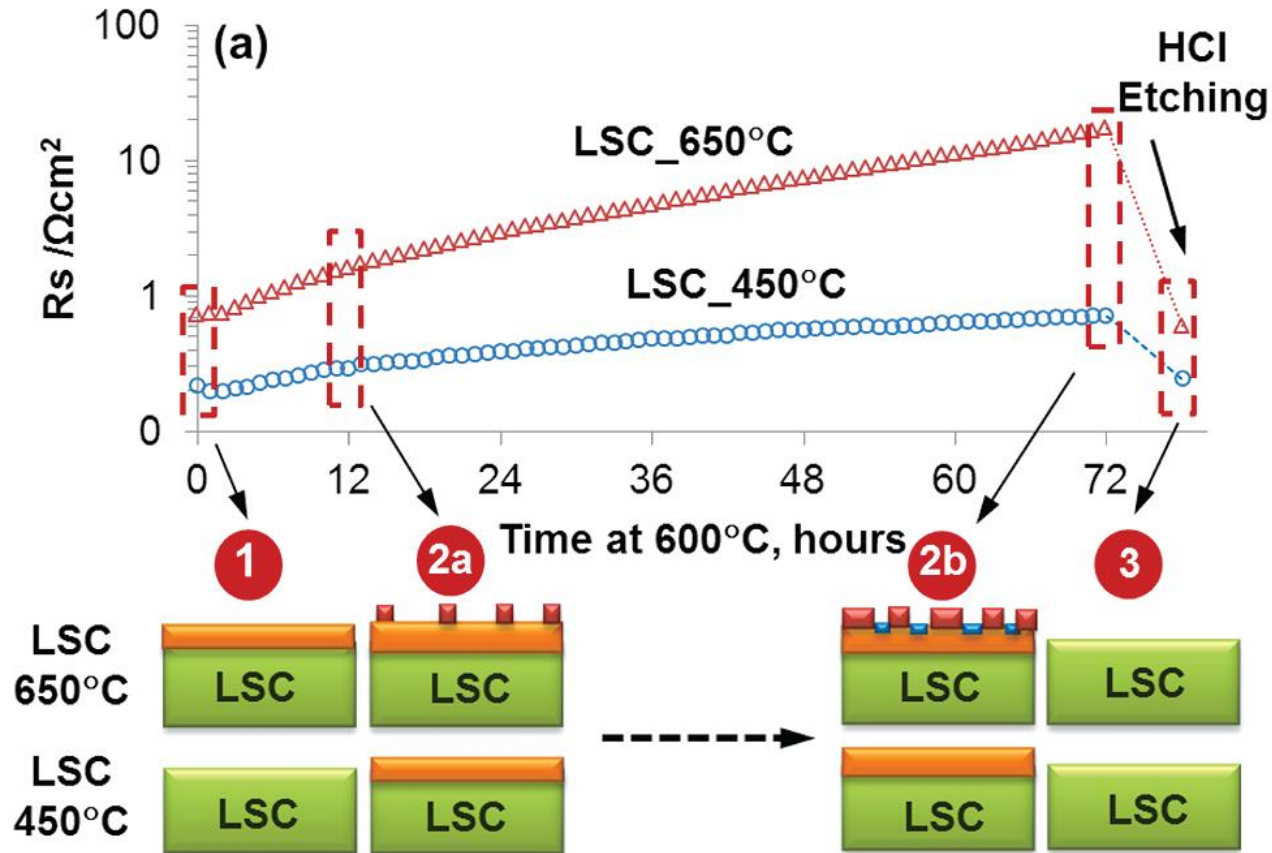
Cai et al., *Chem. Mater.* **24** (2012).

□ Formation of (La,Sr)<sub>2</sub>MnO<sub>4</sub> [3]



Dulli et al., *Phys. Rev. Lett.* **62** (2000).

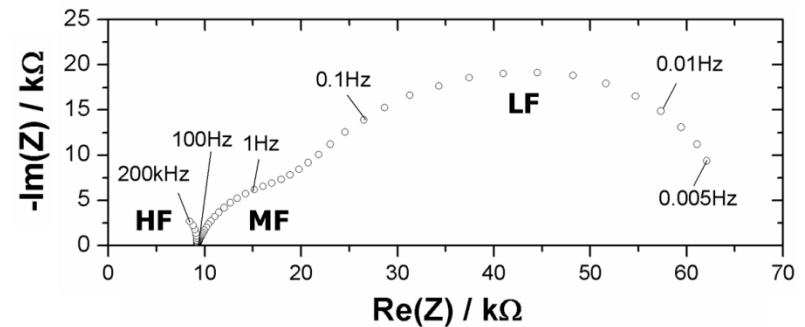
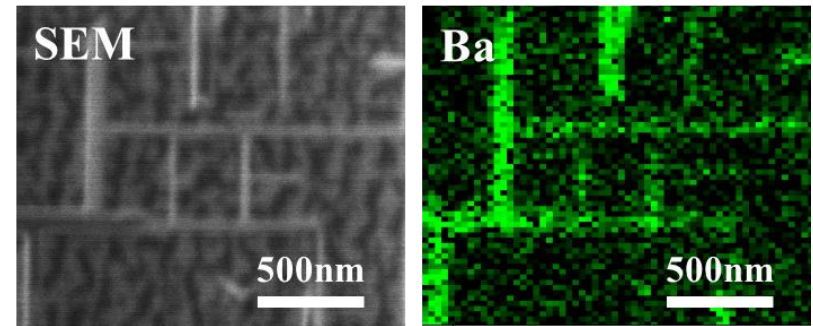
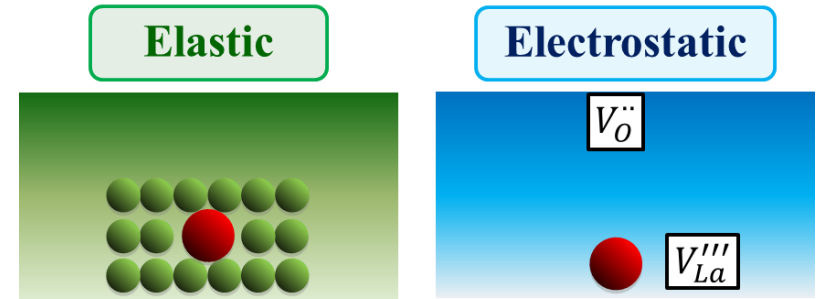
# Surface Chemistry Strongly Affects Surface Activity



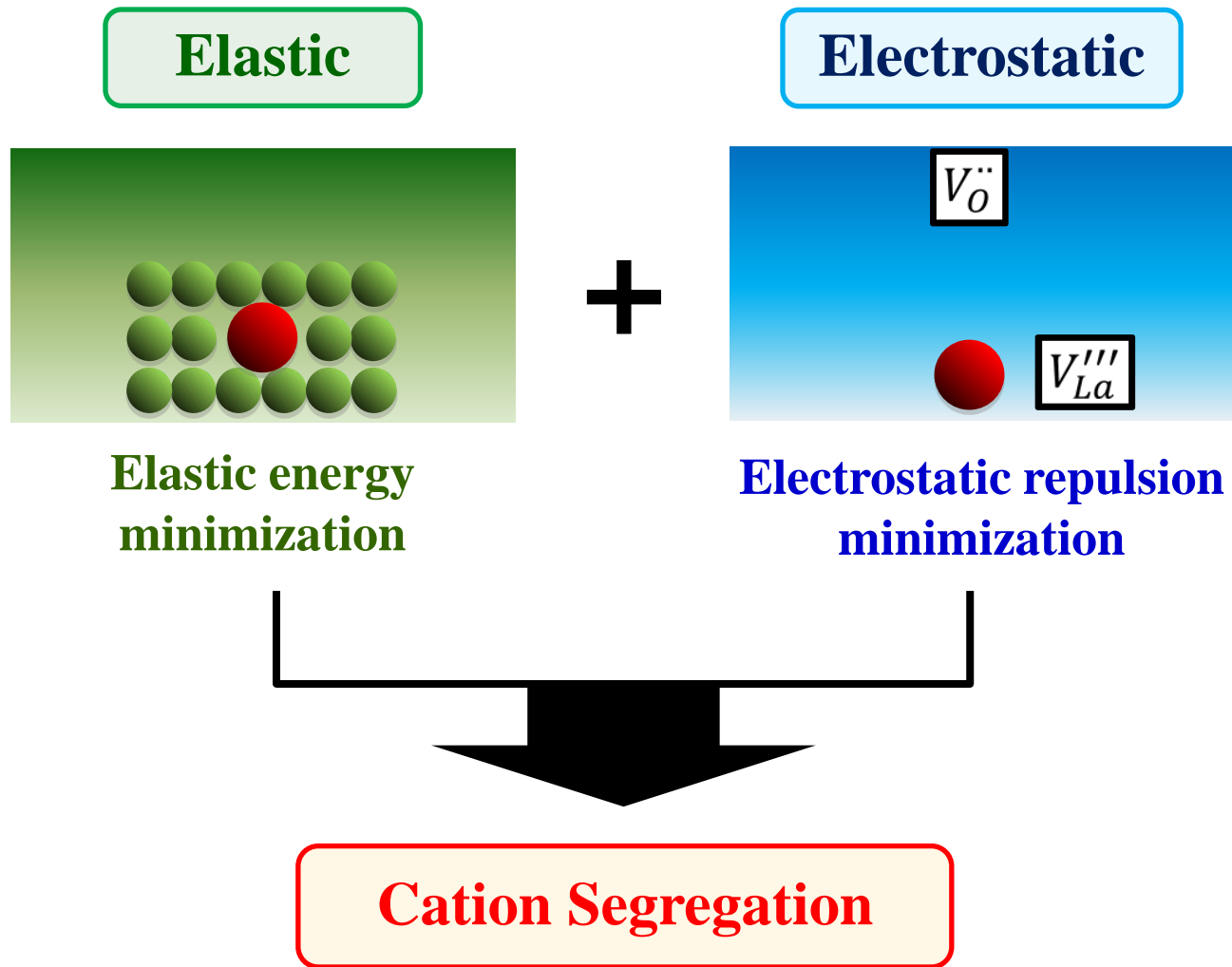
**To control cation segregation for enhanced activity & stability, driving forces must be quantitatively understood**

# Objective

- Quantitatively assess the *key driving forces* of cation segregation on perovskite oxide surfaces
- Determine the *chemical composition* and *structure* of secondary phases on the surface upon cation enrichment
- Assess the effects on *electrochemical* activity



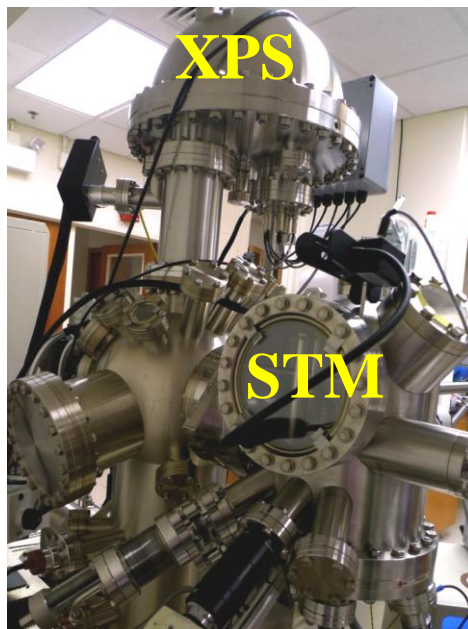
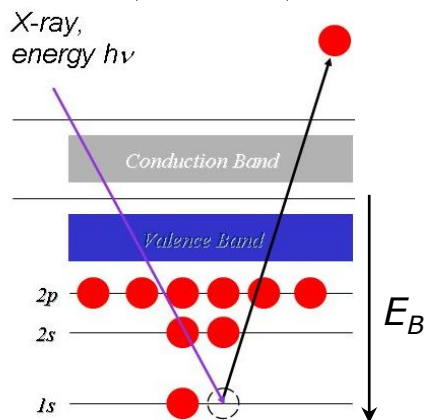
# Hypothesis for Cation Segregation in Perovskites



# Approach

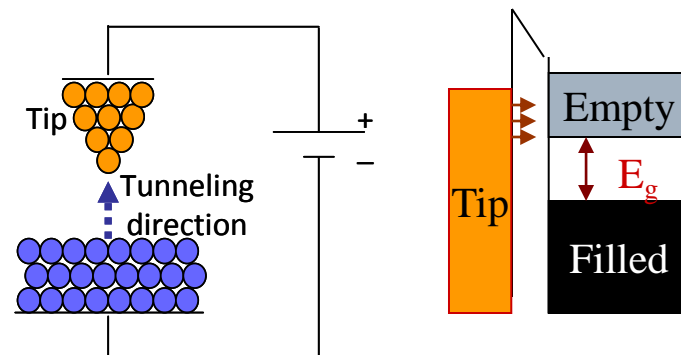
## Surface Chemical State

X-ray Photoelectron Spectroscopy (XPS)  
Nanoscale Auger Electron Spectroscopy (n-AES)

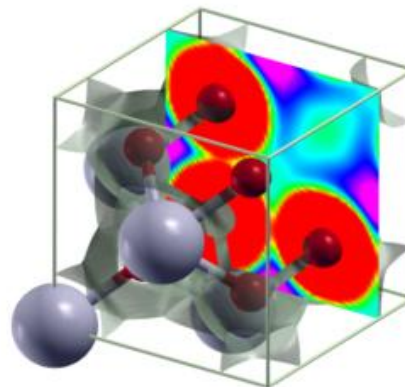


## Surface Electronic Structure

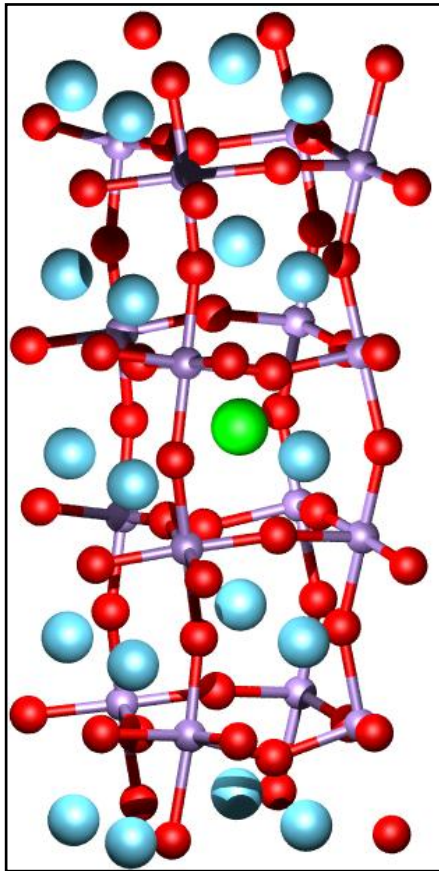
Scanning Tunneling Microscopy / Spectroscopy (STM/STS), *in situ*



**Mechanisms, Energetics and Kinetics of Cation Segregation**  
Electronic structure (DFT+U)



# Control of Elastic Interactions



LaMnO<sub>3</sub> or SmMnO<sub>3</sub>

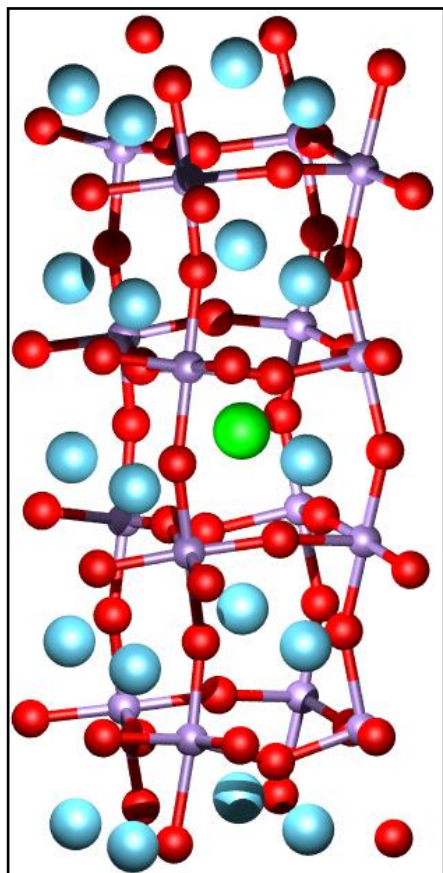
To systematically induce *elastic energy differences*, radius of the selected dopant cations is varied.

Host		Dopant		
		Ca <sup>2+</sup>	Sr <sup>2+</sup>	Ba <sup>2+</sup>
La <sup>3+</sup>	$\Delta r$ (Å)	-0.02	0.08	0.25
	$\Delta r/r_0$ (%)	-1.5	5.9	18.4
Sm <sup>3+</sup>	$\Delta r$ (Å)	0.10	0.20	0.37
	$\Delta r/r_0$ (%)	8.1	16.1	29.8

Increasing *dopant size* relative to the host cation



# Control of Electrostatic Interactions



LaMnO<sub>3</sub> or SmMnO<sub>3</sub>

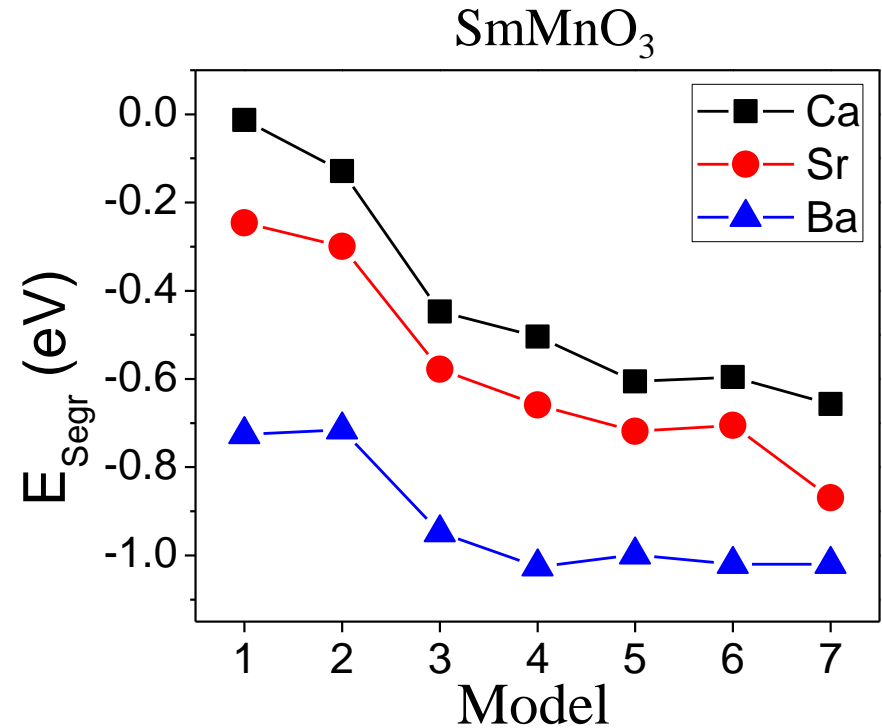
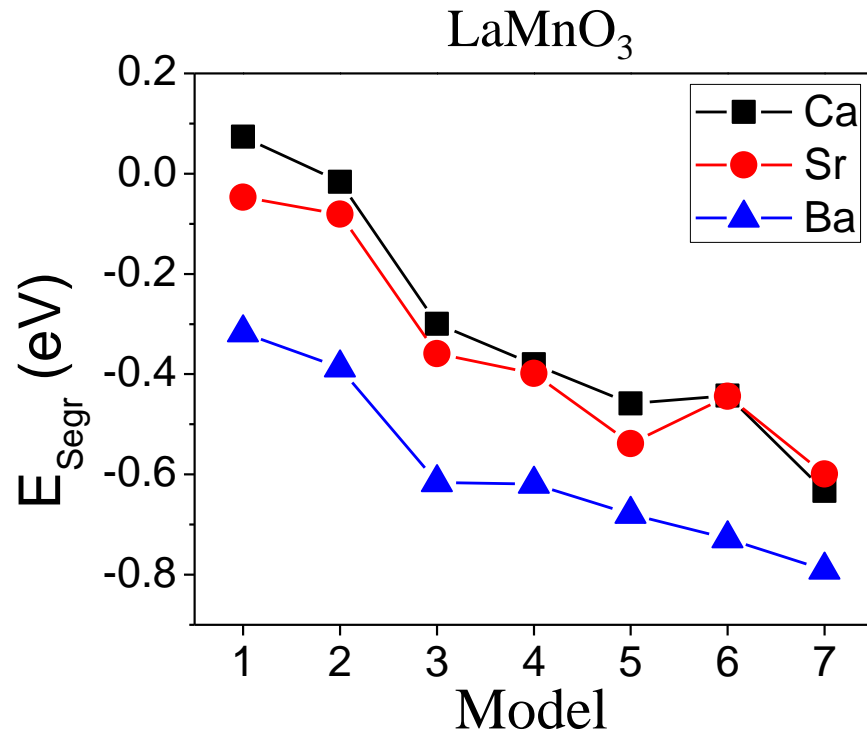
Seven models to represent the variation of *electrostatic interactions* are constructed by controlling the distribution of charged oxygen- and cation-vacancies.

Model	1	2	3	4	5	6	7
Surface	$\bar{V}_{La}$	$\bar{V}_{La}$			$\bar{V}_O$		$\bar{V}_O$
Bulk	$\bar{V}_O$		$\bar{V}_O$			$\bar{V}_{La}$	$\bar{V}_{La}$
	$-$ $+$	$-$	$-$ $+$	$-$	$-$	$-$ $-$	$-$ $-$



Increasing *attractive interaction* to the dopant cation to the surface

# Both Elastic and Electrostatic Drivers are Important



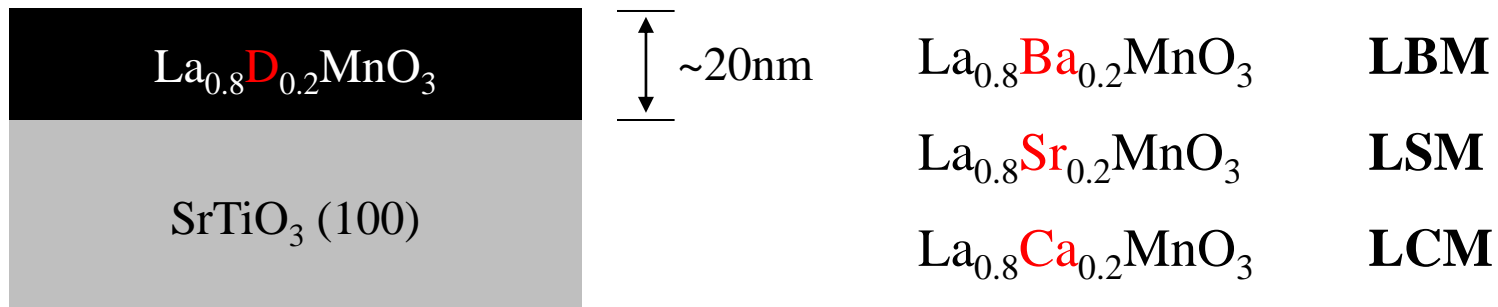
Model	1	2	3	4	5	6	7
Surface	- V <sub>La</sub>	- V <sub>La</sub>			+ V <sub>O</sub>		+ V <sub>O</sub>
Bulk	● - + V <sub>O</sub>	● - V <sub>O</sub>	● - + V <sub>O</sub>	● - V <sub>O</sub>	● - V <sub>O</sub>	● - - V <sub>La</sub>	● - - V <sub>La</sub>

Han and Yildiz,  
*in preparation.*

# Elastic Interaction: Dopant Size Mismatch

To systematically induce elastic energy differences, radius of the selected dopant cations is varied.

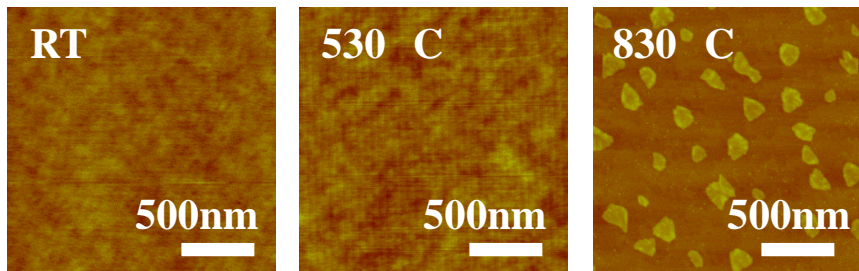
Host cation	Size mismatch	Dopant		
		Ca <sup>2+</sup>	Sr <sup>2+</sup>	Ba <sup>2+</sup>
La <sup>3+</sup>	$\Delta r (= r_{\text{host}} - r_{\text{dopant}})$ (Å)	-0.02	0.08	+0.25
	$\Delta r/r_0$ (%)	-1.5	+5.9	+18.4



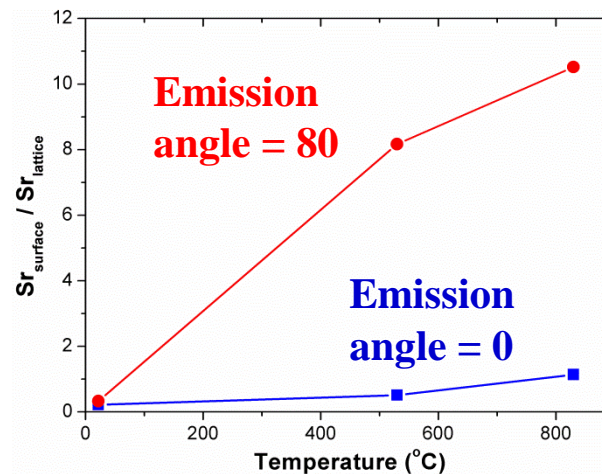
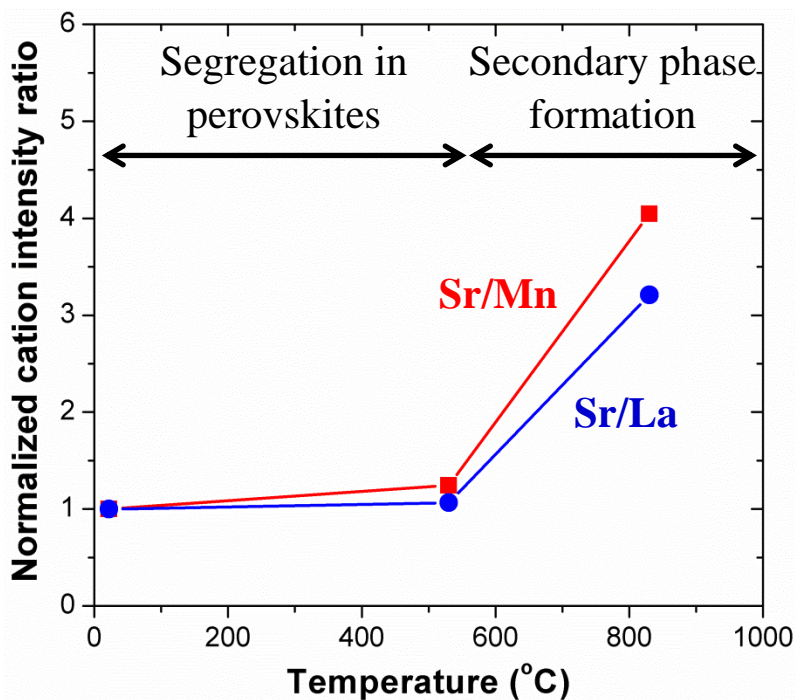
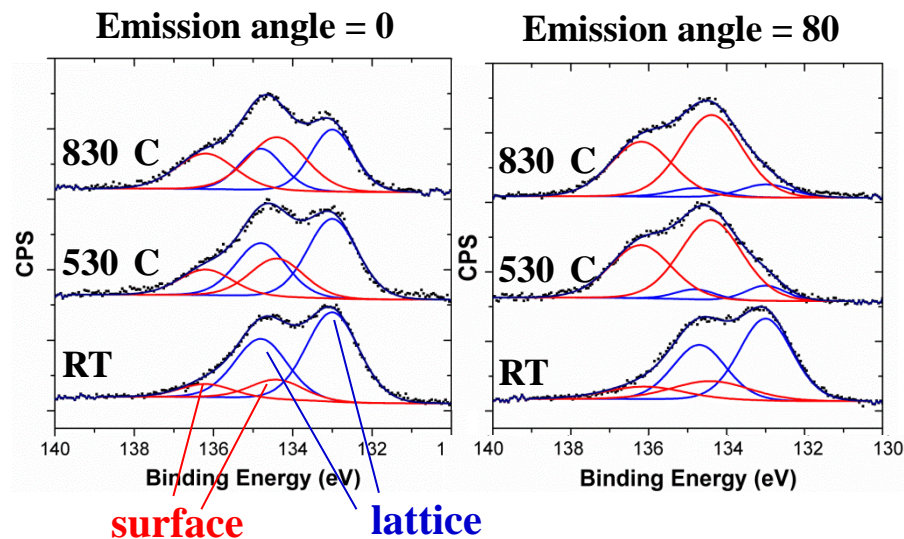
Pulse laser deposition ( $T = 815$  °C in  $p\text{O}_2 = 10$  mTorr)

# Annealing Induced Changes

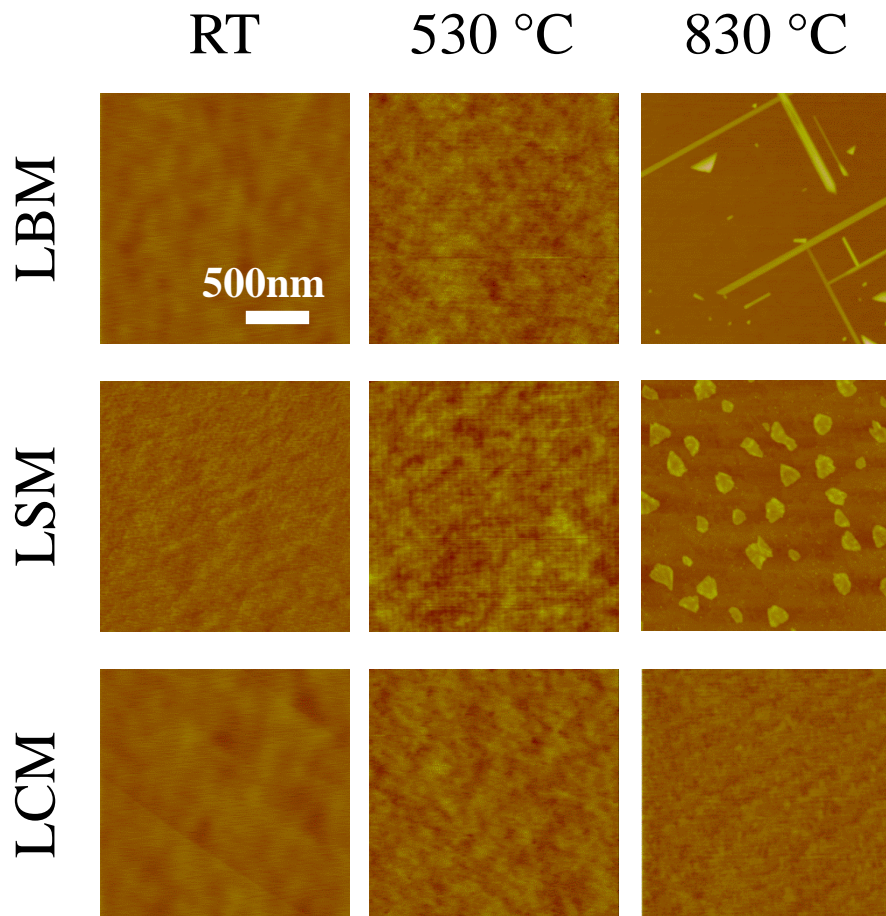
## Structural changes (AFM)



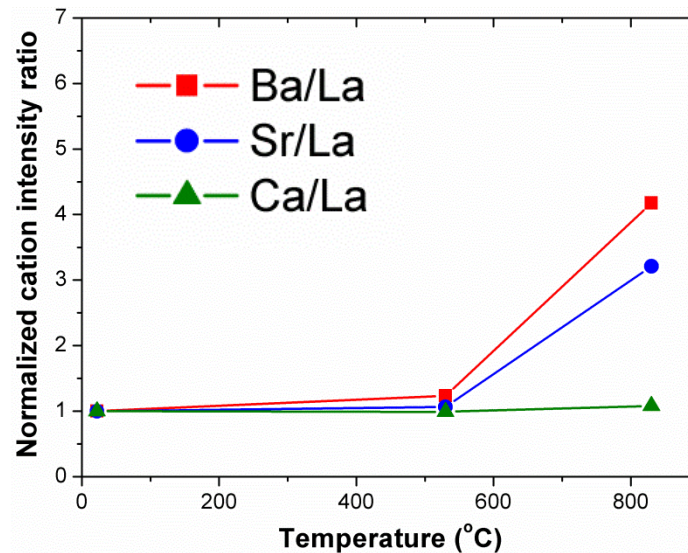
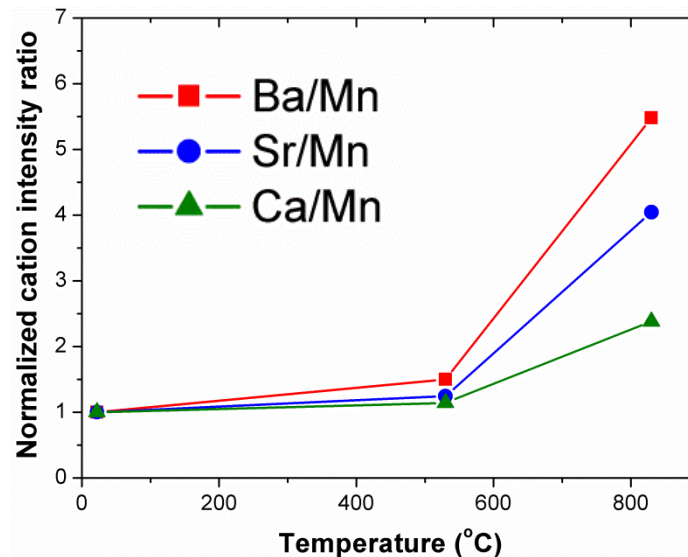
## Chemical changes (Sr 3d)



# Effects of Size Mismatch on Dopant Segregation

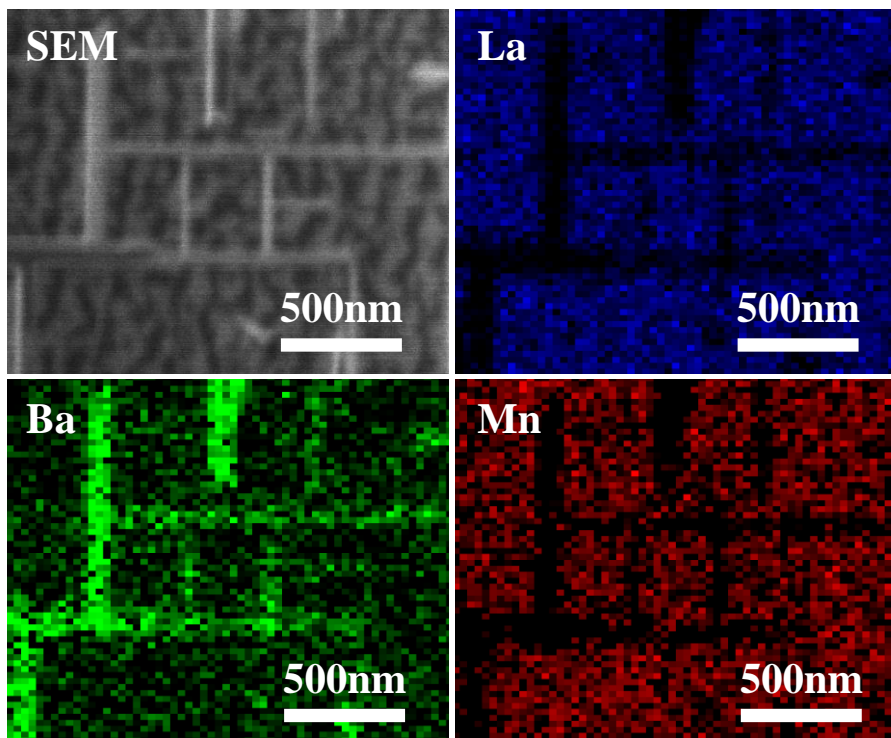


Annealing in air for 1 hr

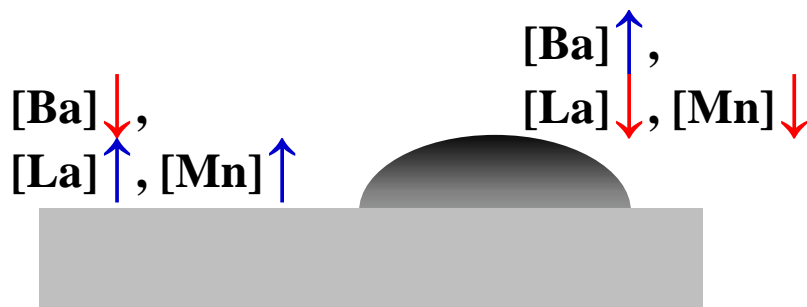
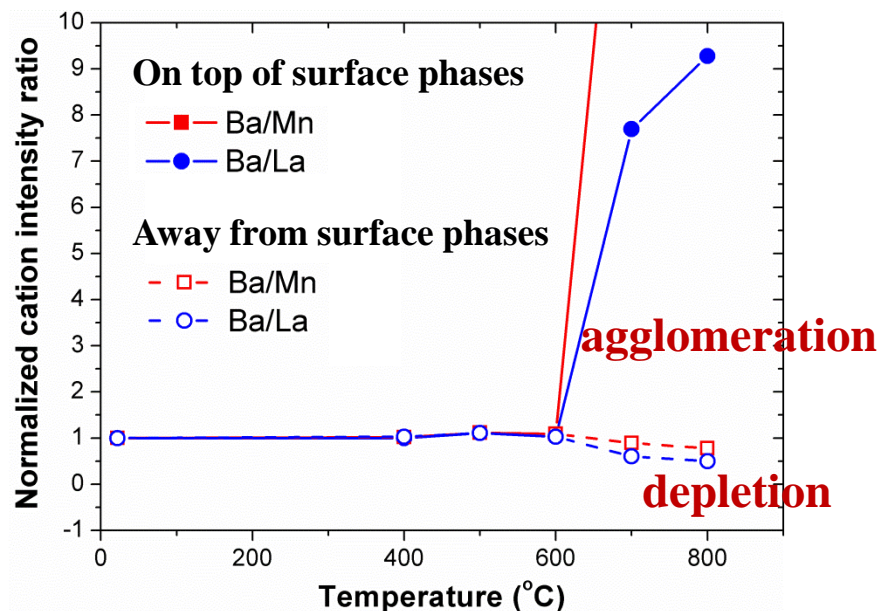


# Localized Chemical Identification

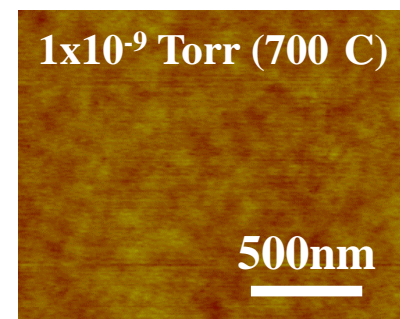
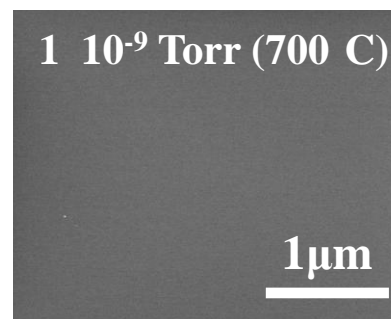
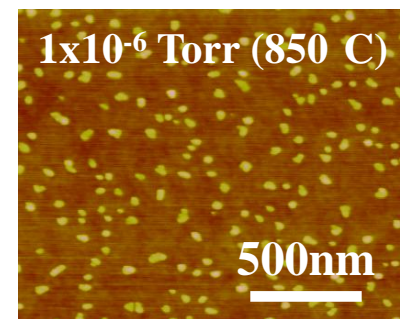
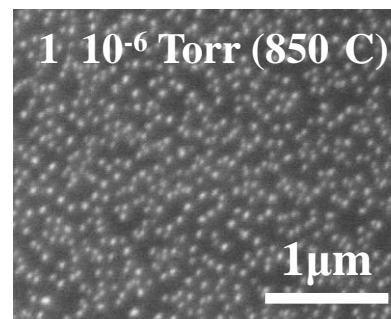
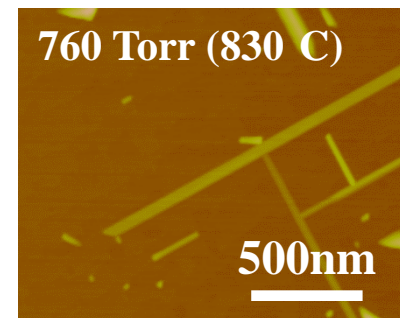
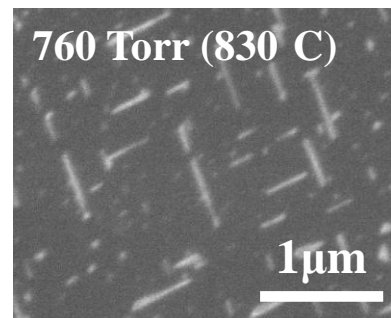
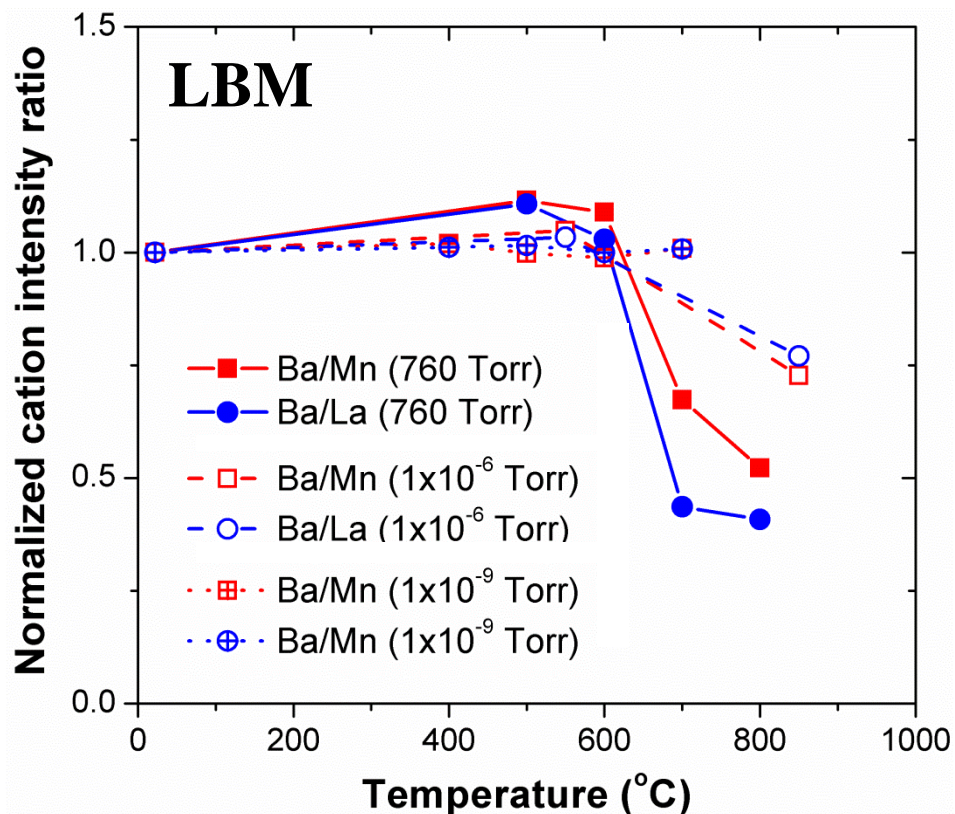
LBM after annealing at 830 °C



Auger electron spectroscopy:  
elemental mapping (left) and point  
spectra (right)

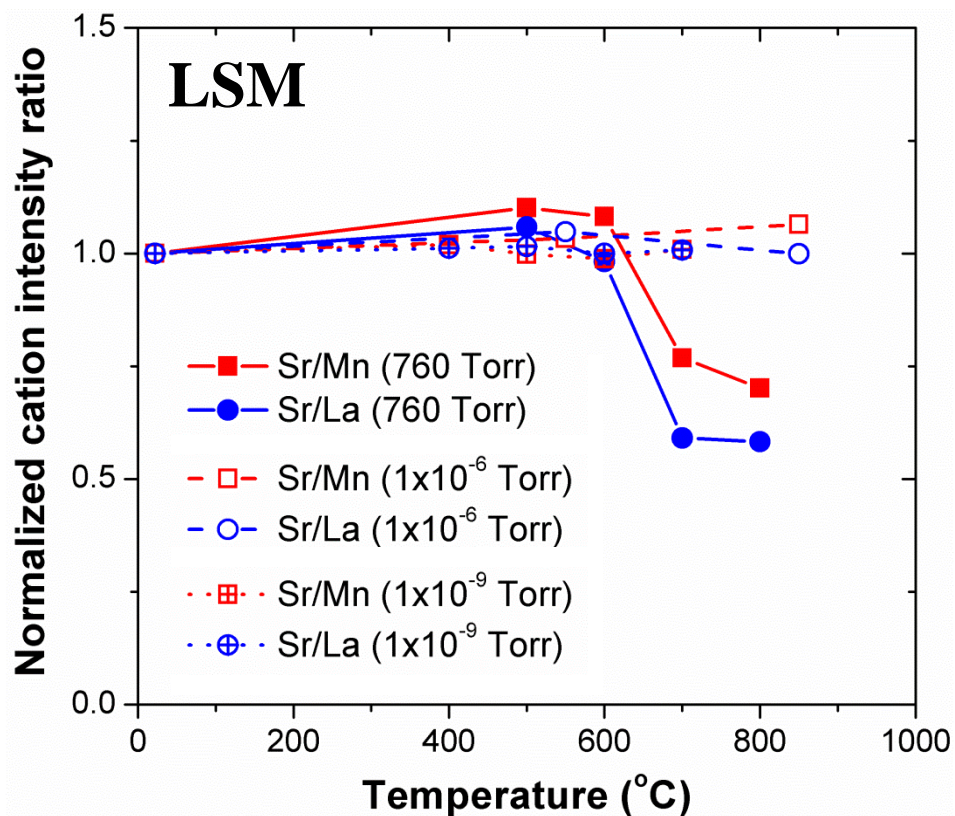


# Effects of Oxygen Pressure on Dopant Segregation

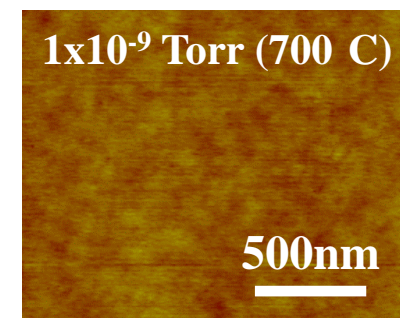
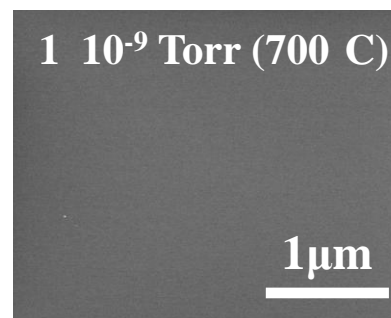
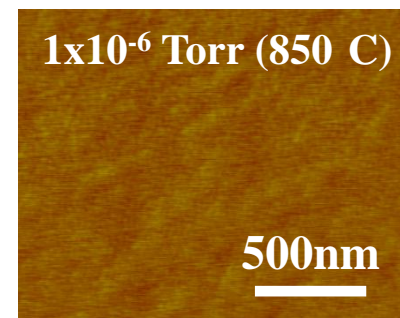
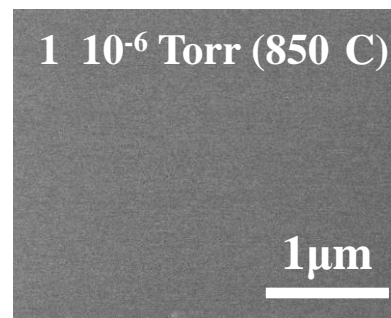
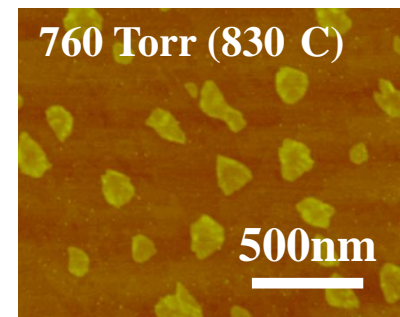
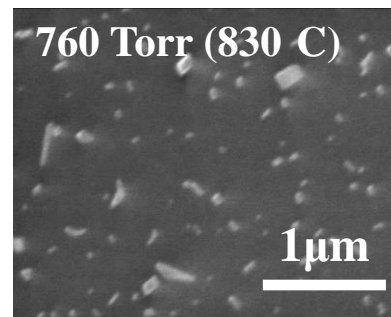


Cation intensity ratio away from the surface phases (background data was used if no features exist)

# Effects of Oxygen Pressure on Dopant Segregation

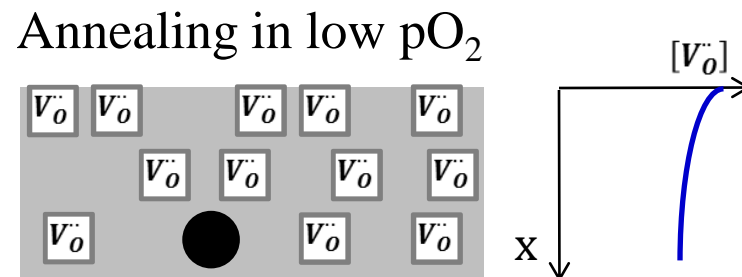
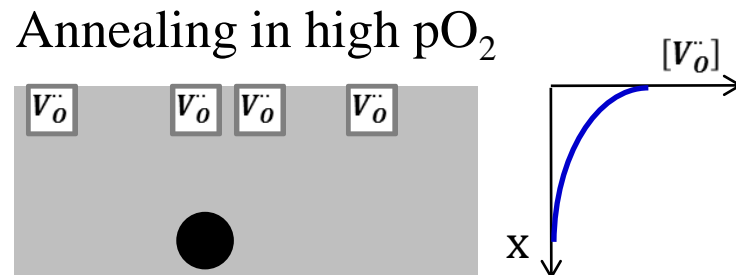
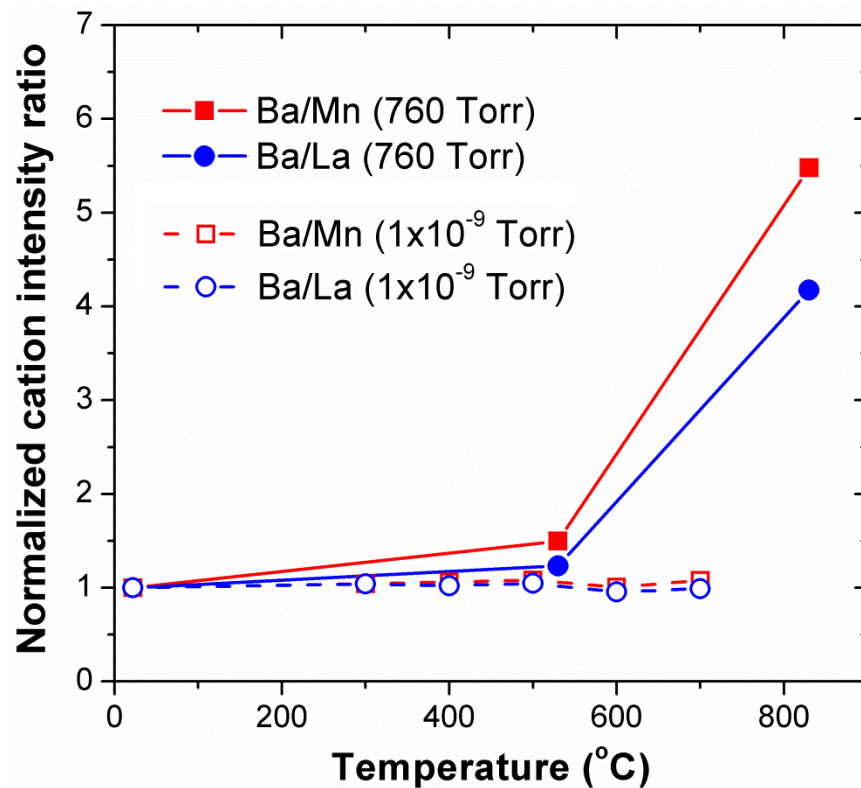


Cation intensity ratio away from the surface phases (background if no features exist)



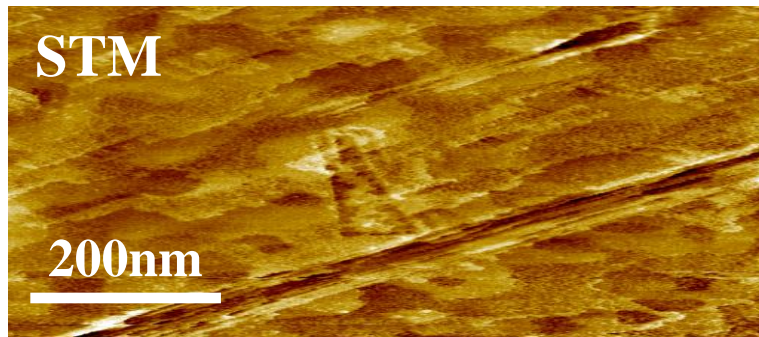
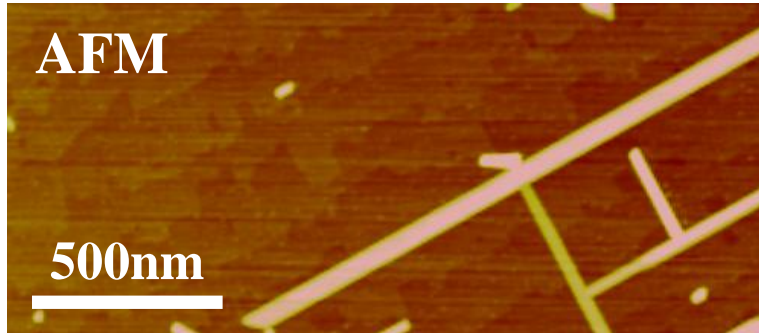


# Effects of Oxygen Pressure on Dopant Segregation

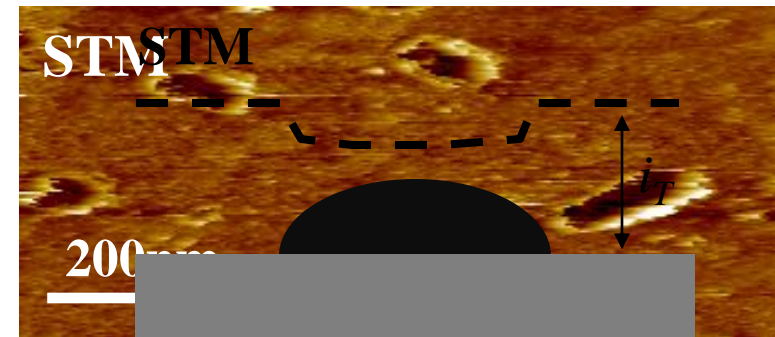
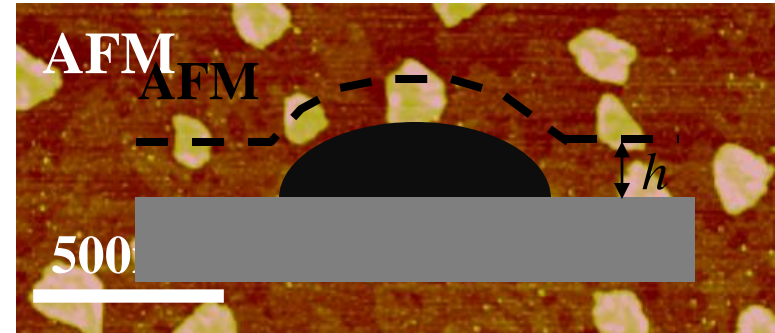


# Electronic Properties of Surface Phases

LBM annealed at 830 °C in air



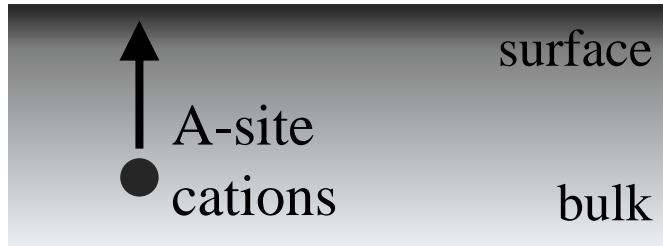
LSM annealed at 830 °C in air



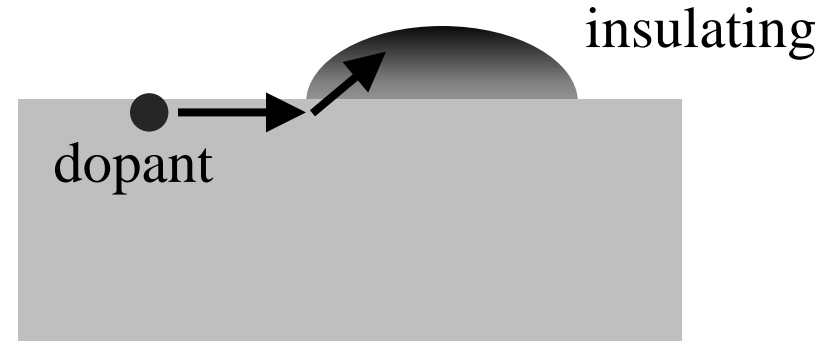
Annealing-induced surface phases are *insulating*

( $E_{g,LBM}$ ,  $E_{g,LSM} \sim 2$  eV,  $E_{g,BaO} \sim 4.5$  eV,  $E_{g,SrO} \sim 5.7$  eV)

# Summary

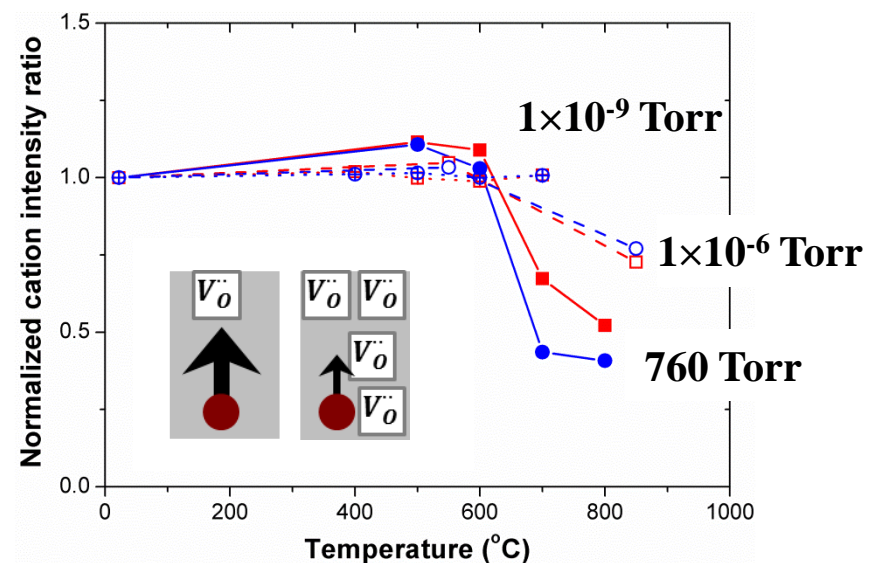
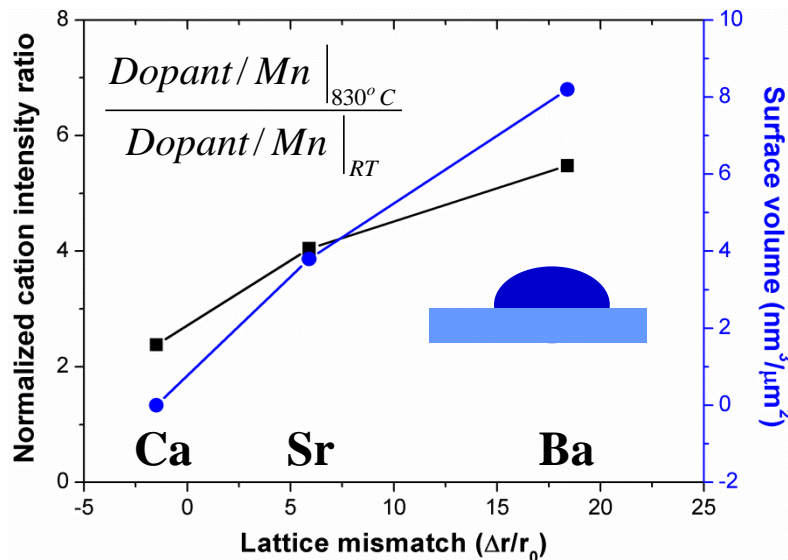


Surface segregation



Agglomeration

Segregation = **Elastic** + **Electrostatic**



# Future Work

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- ✓ Surface *electronic* structures using scanning tunneling microscopy / spectroscopy.
- ✓ *Electrochemical* properties using impedance spectroscopy.

# Acknowledgements

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- ✓ US-DOE, Office of Fossil Energy for financial support.  
(Grant No. DE-NT0004117)
- ✓ Prof. C. Ross and Prof. H. L. Tuller at MIT for the use of their PLD system.

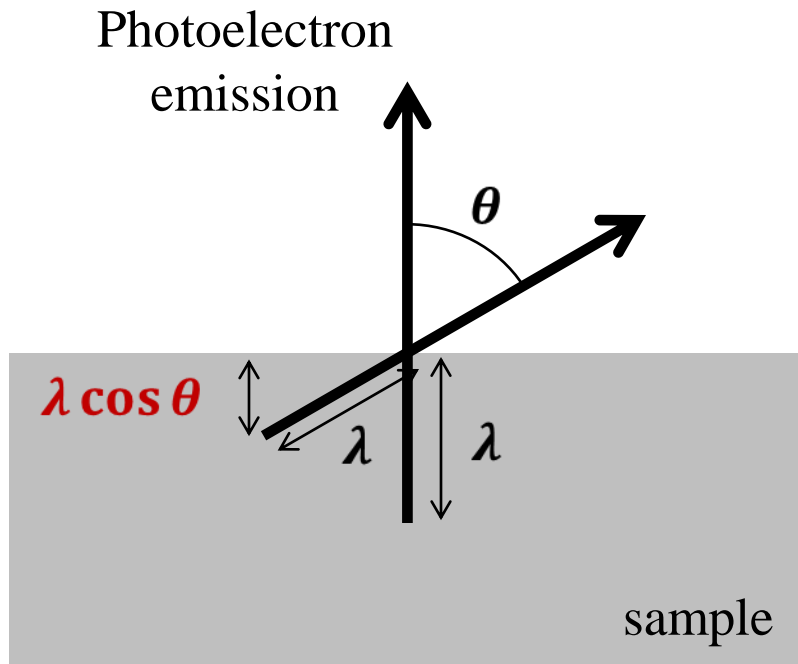
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# Supplementary

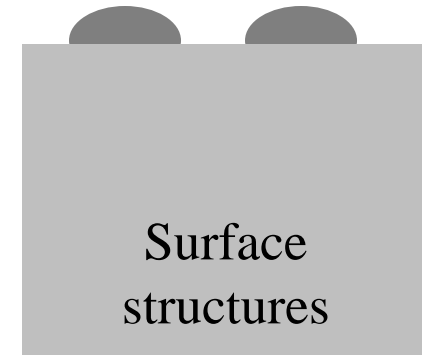
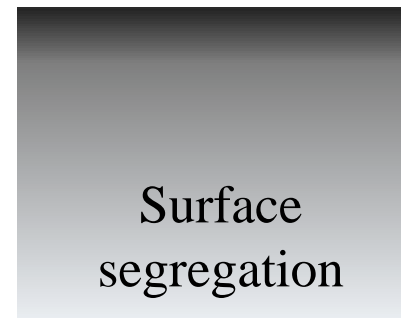
# Angle-Resolved X-ray Photoelectron Spectroscopy

Information depth varies with an emission angle

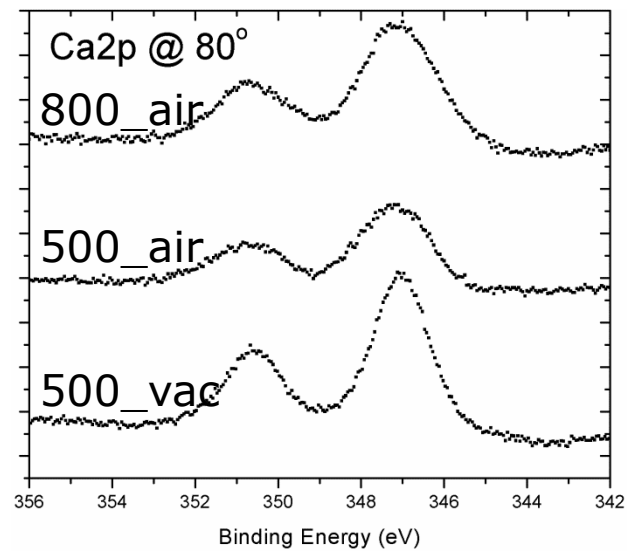
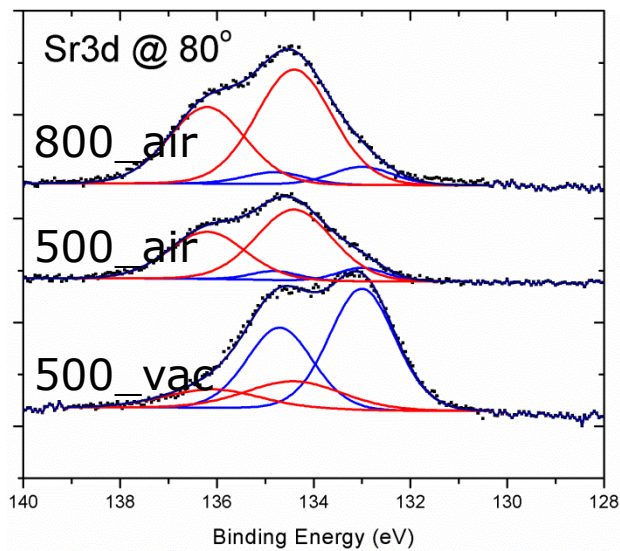
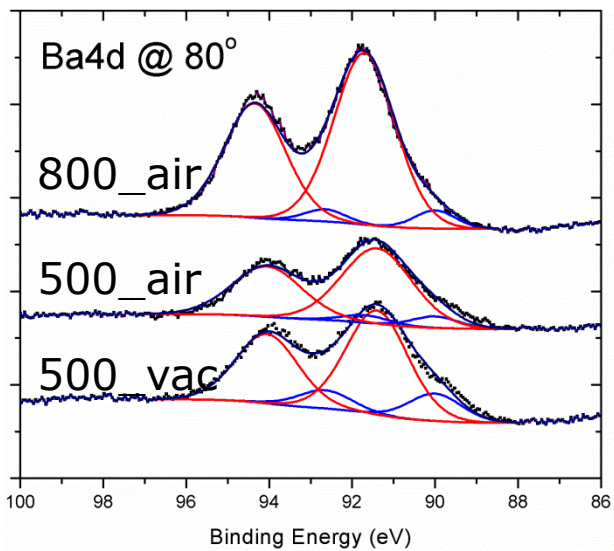
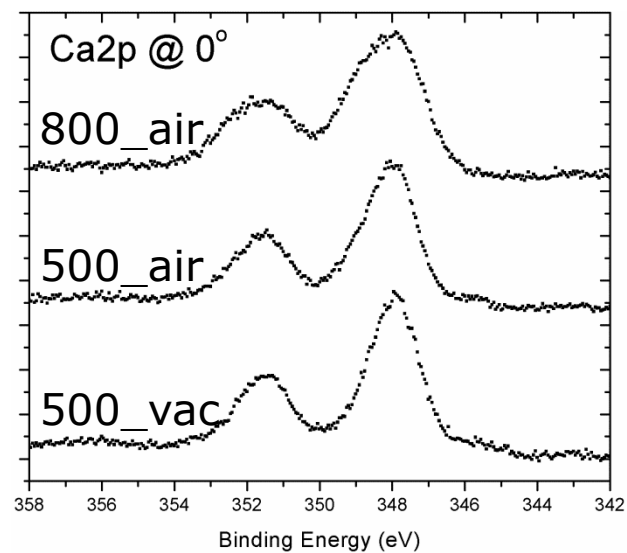
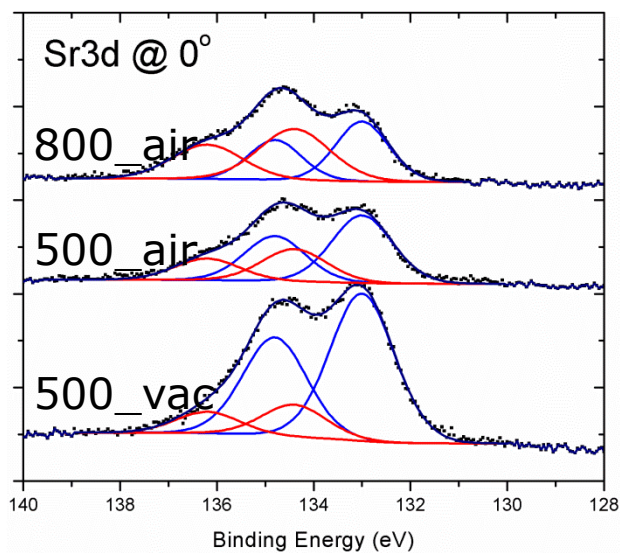
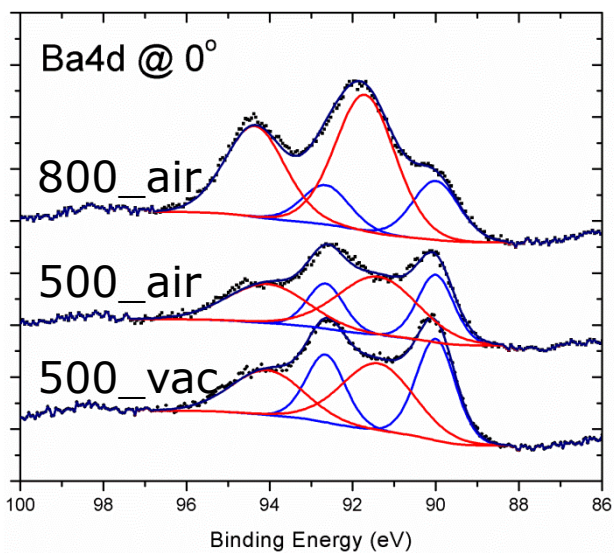
$$I = I_{\infty} \exp(-d/\lambda \cos \theta)$$



Probe the chemical composition  
with a depth information  
(Escape depth of Sr 3d is  $\sim 6\text{nm}$  with  $\theta=0^\circ$ ,  
and  $\sim 1\text{nm}$  with  $\theta=80^\circ$ )



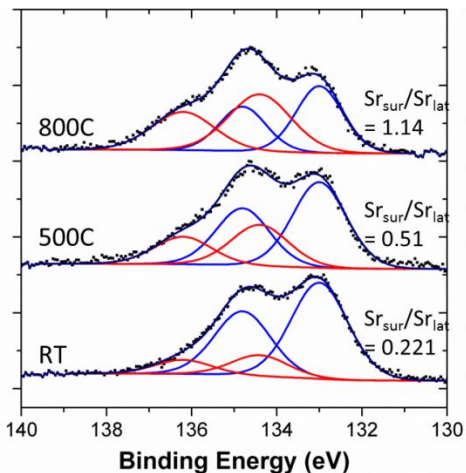
# HR XPS for dopant peaks



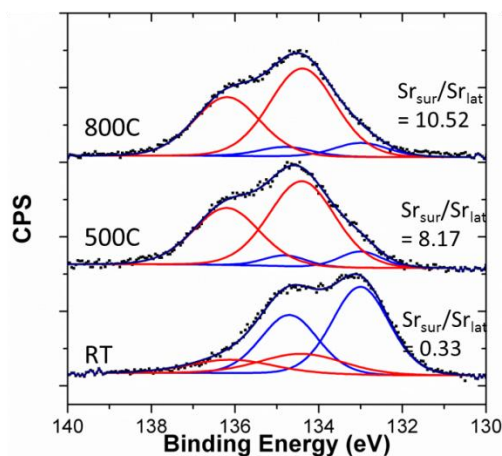


# HR XPS

Sr 3d (theta=0)



Sr 3d (theta=80)

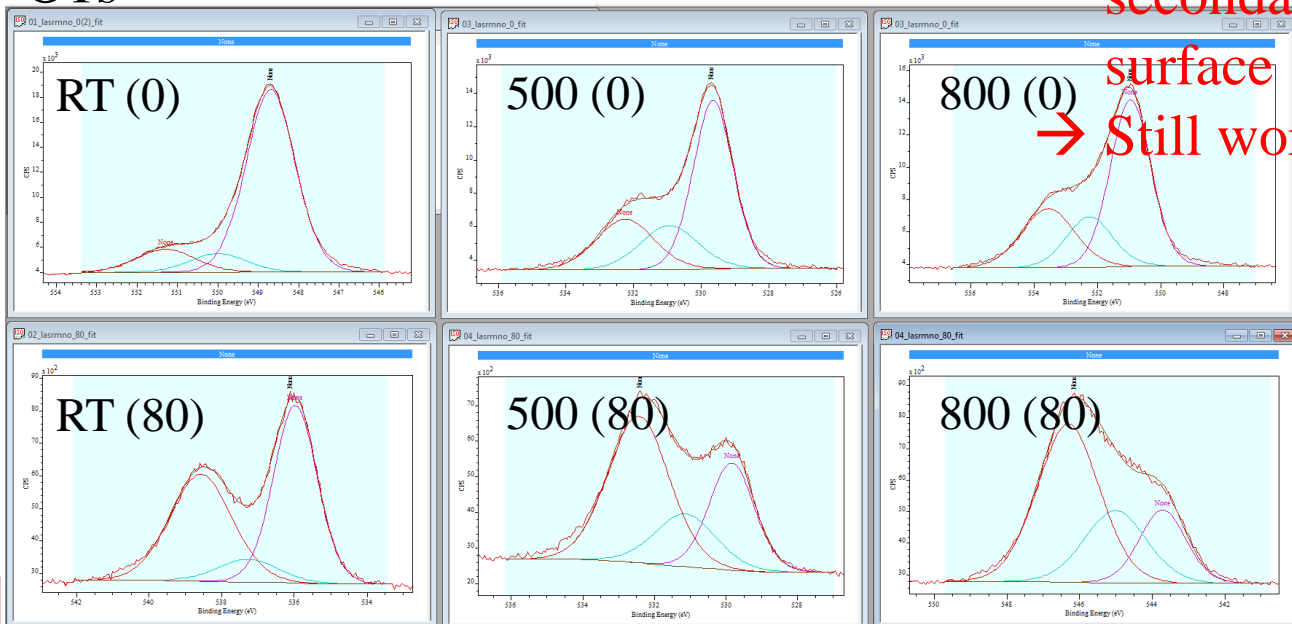


To extract more information from HR XPS, especially to identify the chemical composition of surface layer or secondary phase.

→ This can be useful to describe segregation before forming secondary phases on the surface

→ Still working on this analysis

## O1s



# Effects of Vacancy Distribution on Dopant Segregation

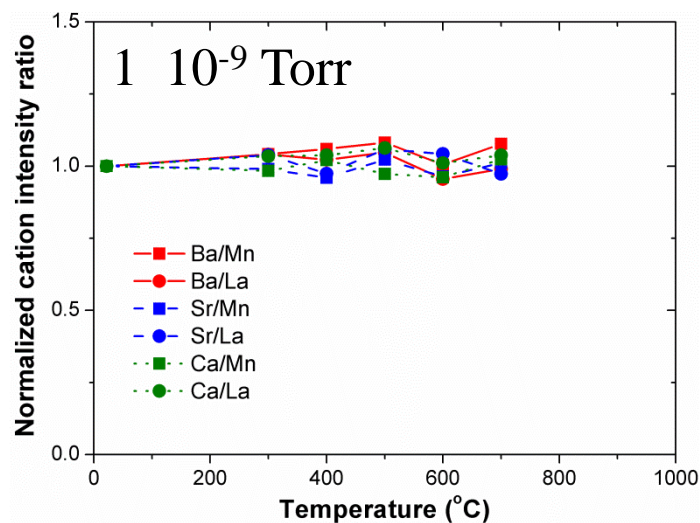
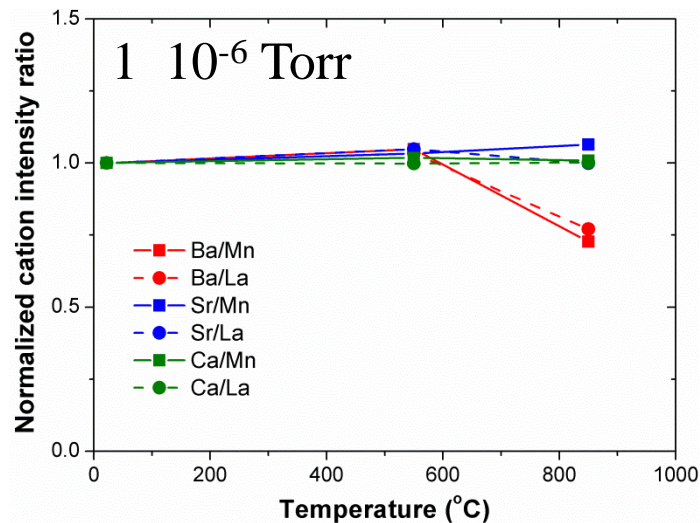
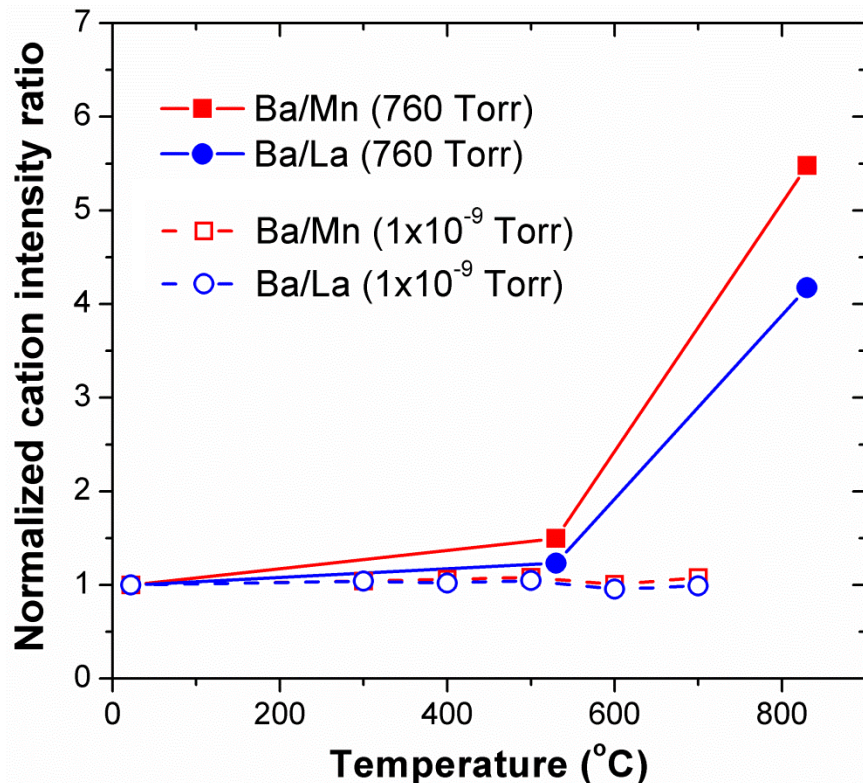
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Other thoughts on this difference (high  $pO_2$  vs. low  $pO_2$ )

If there is strong interaction between the oxygen vacancy (+) and cation vacancy (-), more oxygen vacancies on the surface in UHV would attract more cation vacancies on the surface. Then, it will be difficult for cations to diffuse towards the surface, and/or the cation vacancies will provide rooms for segregated cations?

But, we need to know the diffusivity of cations at 500-800C to see they are mobile enough

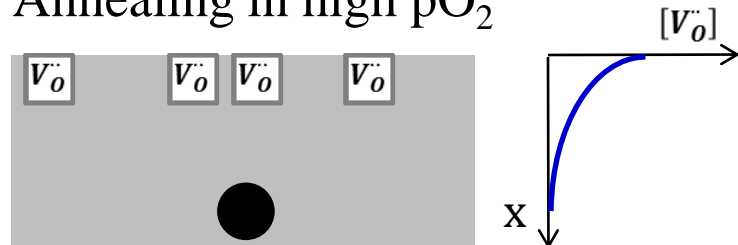
# Effects of Oxygen Pressure on Dopant Segregation



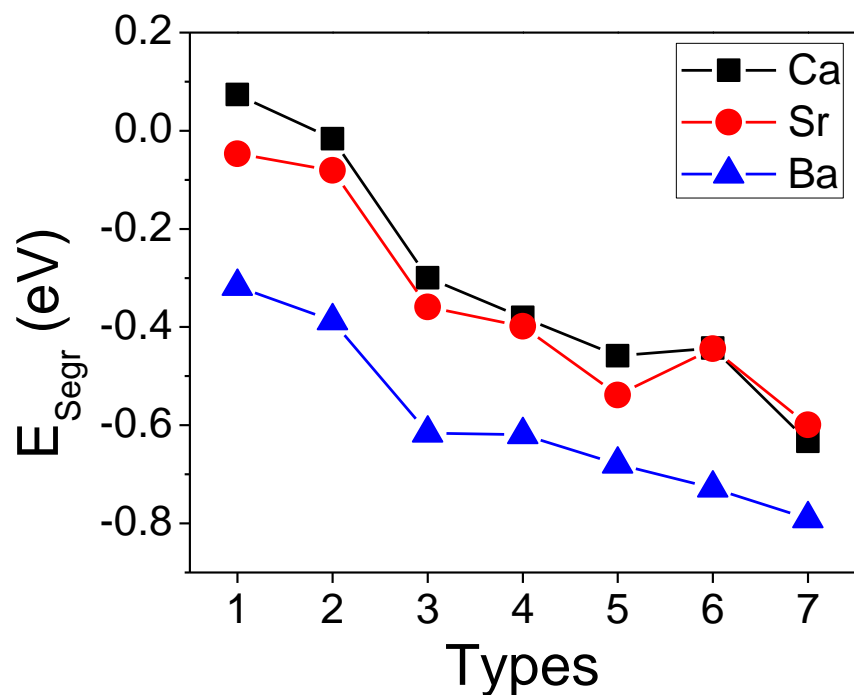
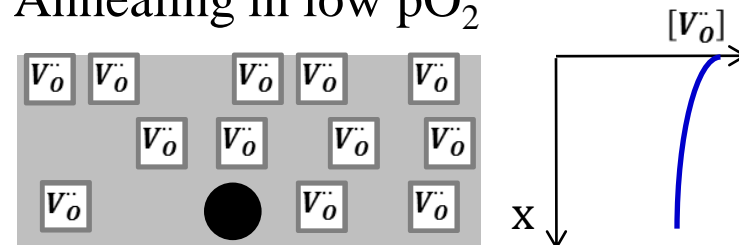
Cation intensity ratio using XPS  
(left) and AES (right)

# Effects of Vacancy Distribution on Dopant Segregation

Annealing in high  $pO_2$



Annealing in low  $pO_2$



Model	1	2	3	4	5	6	7
Surface	$V_{La}^-$	$V_{La}^-$			$V_O^+$		$V_O^+$
Bulk	$V_O^-$	$V_O^-$	$V_O^-$	$V_O^-$	$V_O^-$	$V_{La}^-$	$V_{La}^-$

**Elastic** (dopant size mismatch)  
+ **Electrostatic** (charged defects distribution)